

MADCOR

Irradiation with Low Energy Heavy Ion at
University of Wisconsin and PIE capability

Adrien Couet – Associate
Professor UW-Madison

UW-Madison: Kumar Sridharan,
Hongliang Zhang, Zefeng Yu, Benoit Queylat, Michael
Moorehead, Cody Falconer, Nathan Curtis, Cole Evered, Nick
Crnkovich, Bao-Phong Nguyen, Kim Kriewaldt

FermiLab: Frederique Pellemoine, Kavin Ammigan, Sujit
Bidhar



WISCONSIN
UNIVERSITY OF WISCONSIN-MADISON



UW – ION BEAM LABORATORY (IBL)



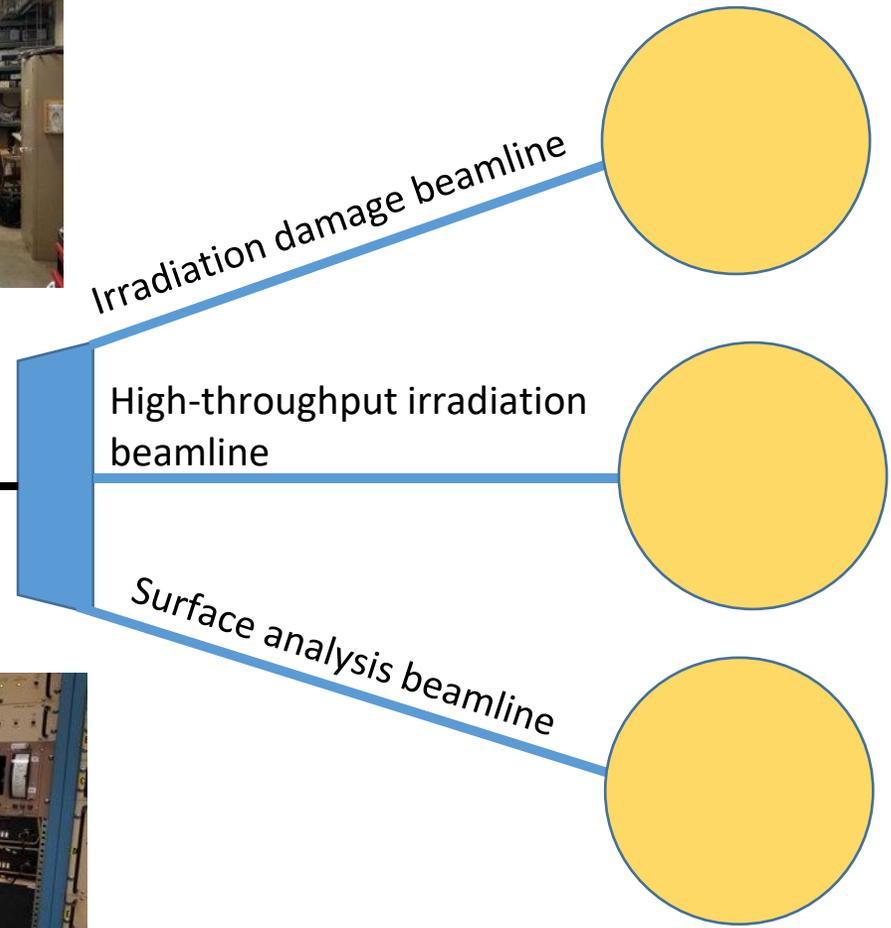
- 1.7 MV tandem accelerator from National Electrostatics Corporation (NEC)
- Temperature monitored with thermocouples and IR camera
- Various samples geometries
- Rastered or defocused beam
- Toroidal Volume Ion Source (TORVIS) and Source of Negative Ions via Cesium Sputtering (SNICS) ion sources



UW – ION BEAM LABORATORY (IBL)



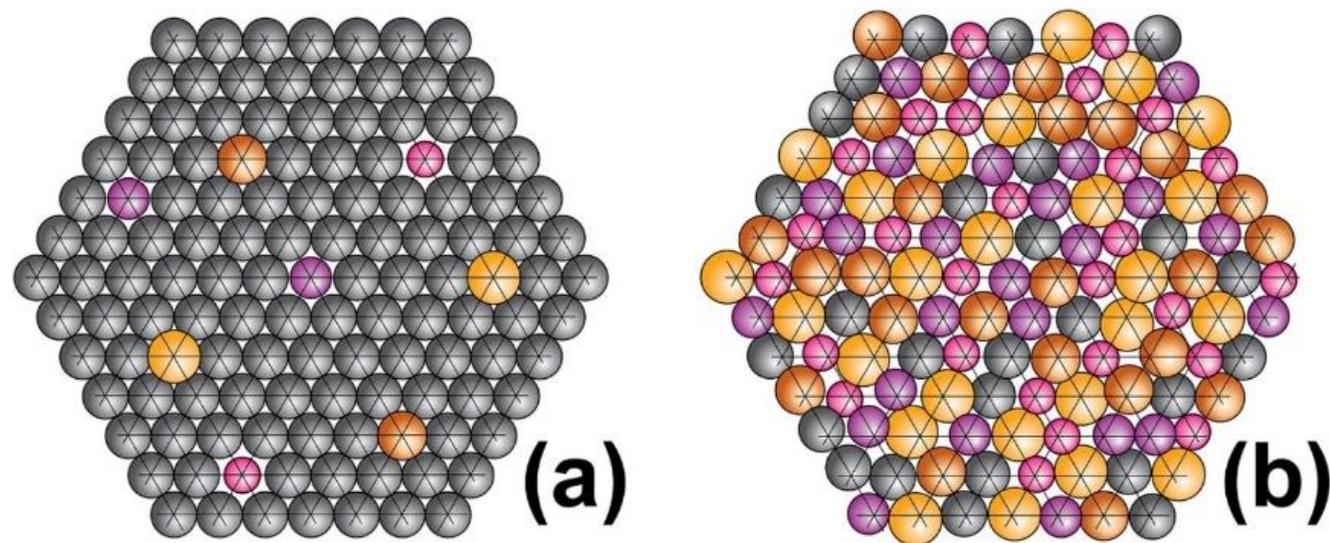
TORVIS: H, D, He
 SNICS: C, Si, Fe, Ni,
 others...



- Water chiller and liquid metal cooling for precise temperature control,
- Heating cartridges (up to 1000C)
- **New installation of high-throughput irradiation beamline (2021/2022)**
- Large chamber with load lock
- 3D moving stage for high-throughput irradiation
- Controlled laser heating from front
- **New installation of ion beam analysis beamline (2022/2023)**
- Rutherford backscattering spectrometry (RBS)
- Particle-induced X-ray emission (PIXE).

HIGH-ENTROPY ALLOY (HEA) OVERVIEW

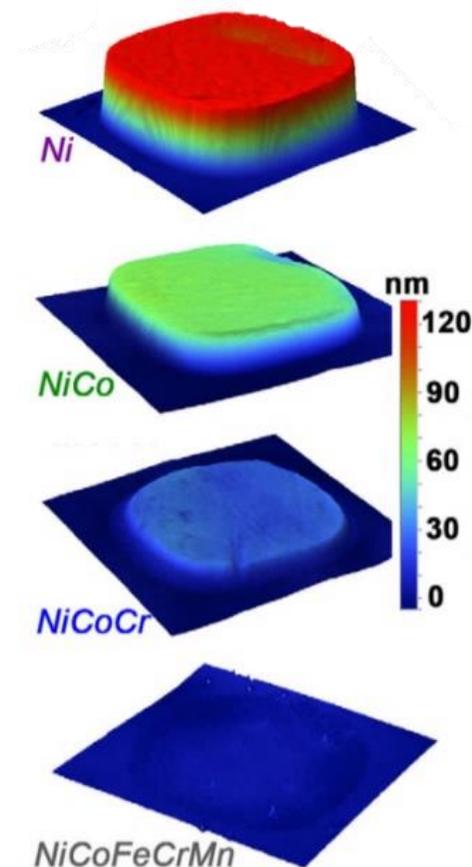
- Develop robust beam-intercepting devices (targets, beam windows, beam dumps, collimators...) to maximize physics benefits of future experiments
- Better assessment of current materials used in accelerators
- Develop novel materials to improve lifetime of components under extreme conditions → new materials specifically addressing the requirements of future target facilities.



(a) Conventional alloy, (b) High-entropy alloy (Miracle, Senkov | 2017)

HIGH-ENTROPY ALLOY (HEA) OVERVIEW

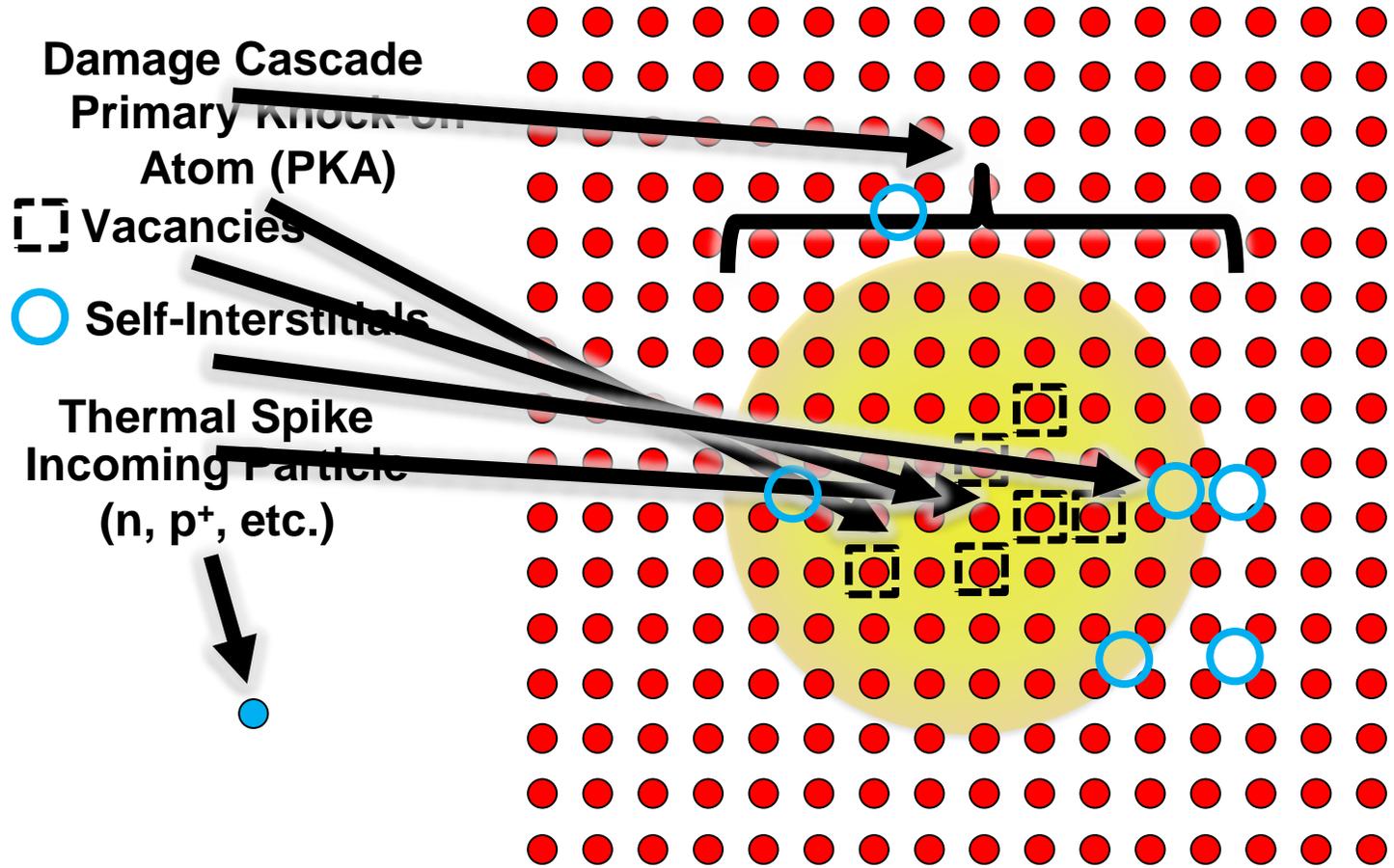
- High-entropy alloys (HEAs) are alloys with multiple principal elements
- Typically, no element >35 at%
- Usually defined as being primarily a solid-solution matrix
- Promising properties have been observed:
 - High-temperature strength
 - High specific strength
 - Enhanced radiation tolerance



Swelling of increasingly complex alloys under ion irradiation (Jin | 2016)



RADIATION DAMAGE IN METALS



Long-Term Defect Evolution:

Quench: As time progresses and more damage cascades are produced, the population of atoms are mobile enough to quickly find new lattice sites with sufficient energy to remove the atom from its original site.

Thermal Spike: Following the thermal spike, displaced atoms are mobile enough to quickly find new lattice sites with sufficient energy to remove the atom from its original site.

Primary Collision: As time progresses and more damage cascades are produced, the population of atoms are mobile enough to quickly find new lattice sites with sufficient energy to remove the atom from its original site.

However, as heat from the thermal spike dissipates, the mobility of atoms is reduced. Point defects can diffuse towards free surfaces or interfaces (grains) or dislocations, leaving severe primary knock-on defects. Point interstitial atoms and vacancies can also recombine, forming a thermal spike voids and dislocations.



RADIATION TOLERANT MICROSTRUCTURES

Typical approach to improving radiation tolerance: increase sink density

Oxide-dispersion
strengthened (ODS) alloys

Nanocrystalline alloys

Cold-worked alloys

HOWEVER

**These engineered microstructures are not stable under all
temperature and damage regimes**

(a)

100 nm

100 nm

100 nm

(Allen | 2008)

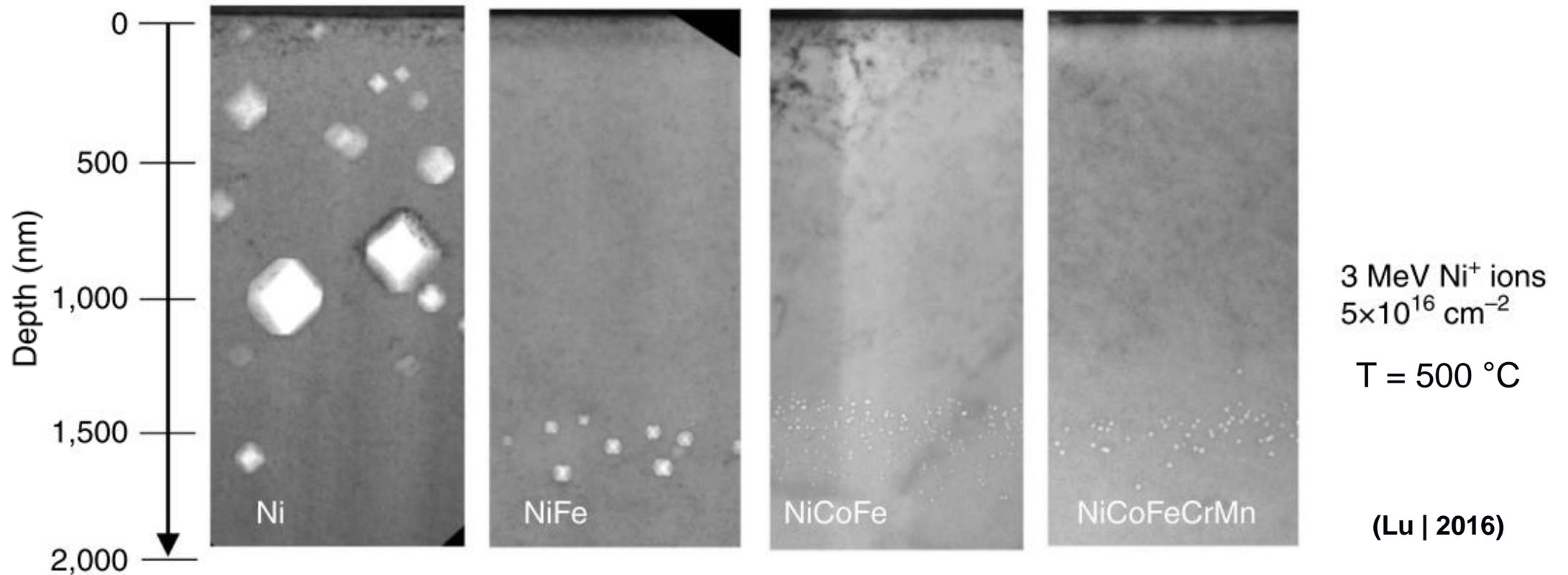
(Barr | 2018)

(Alshater | 2018)



RADIATION TOLERANCE IN HEAs

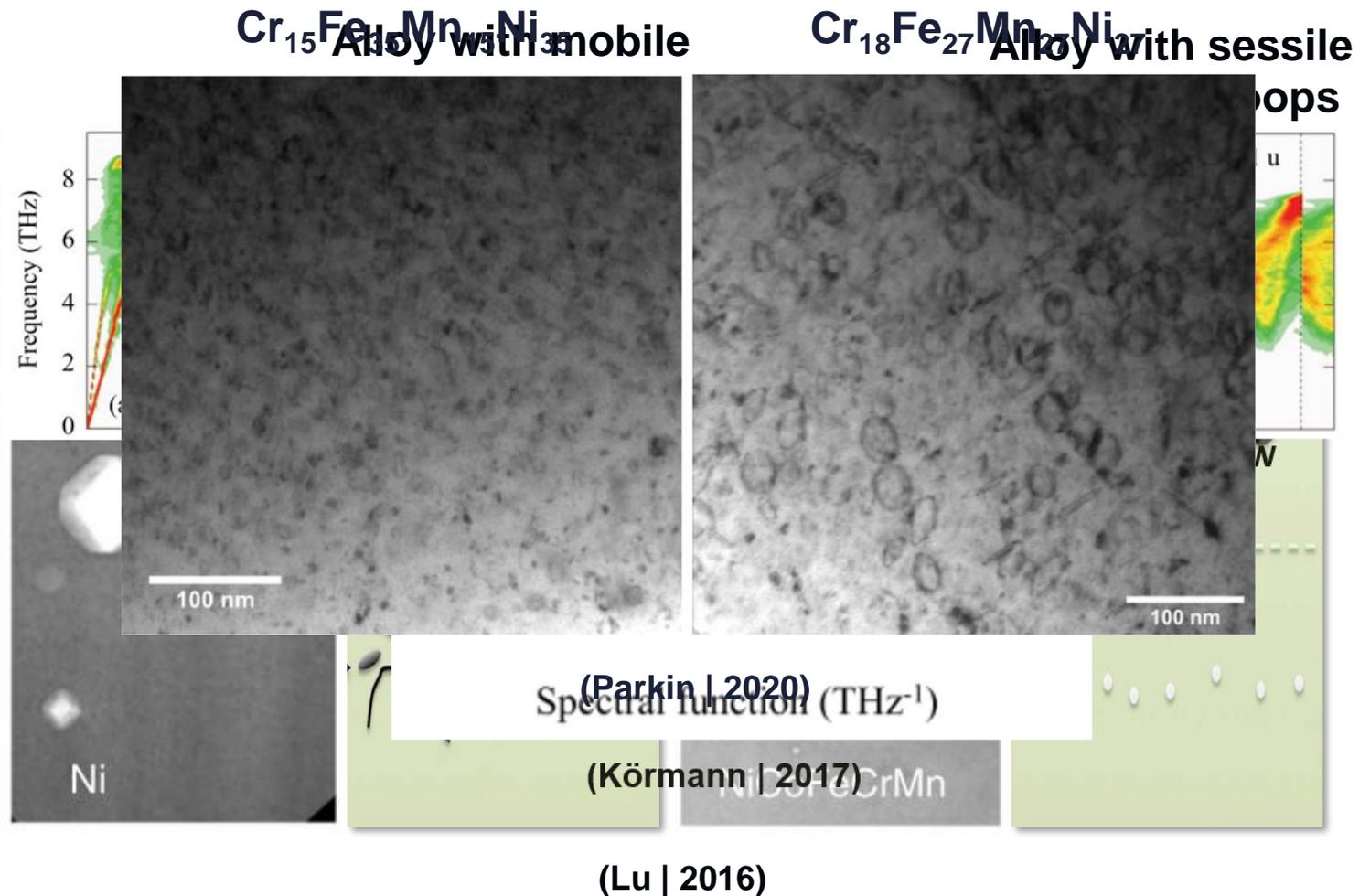
Both modeling and experimentation have shown HEAs can exhibit enhanced radiation tolerance in the matrix



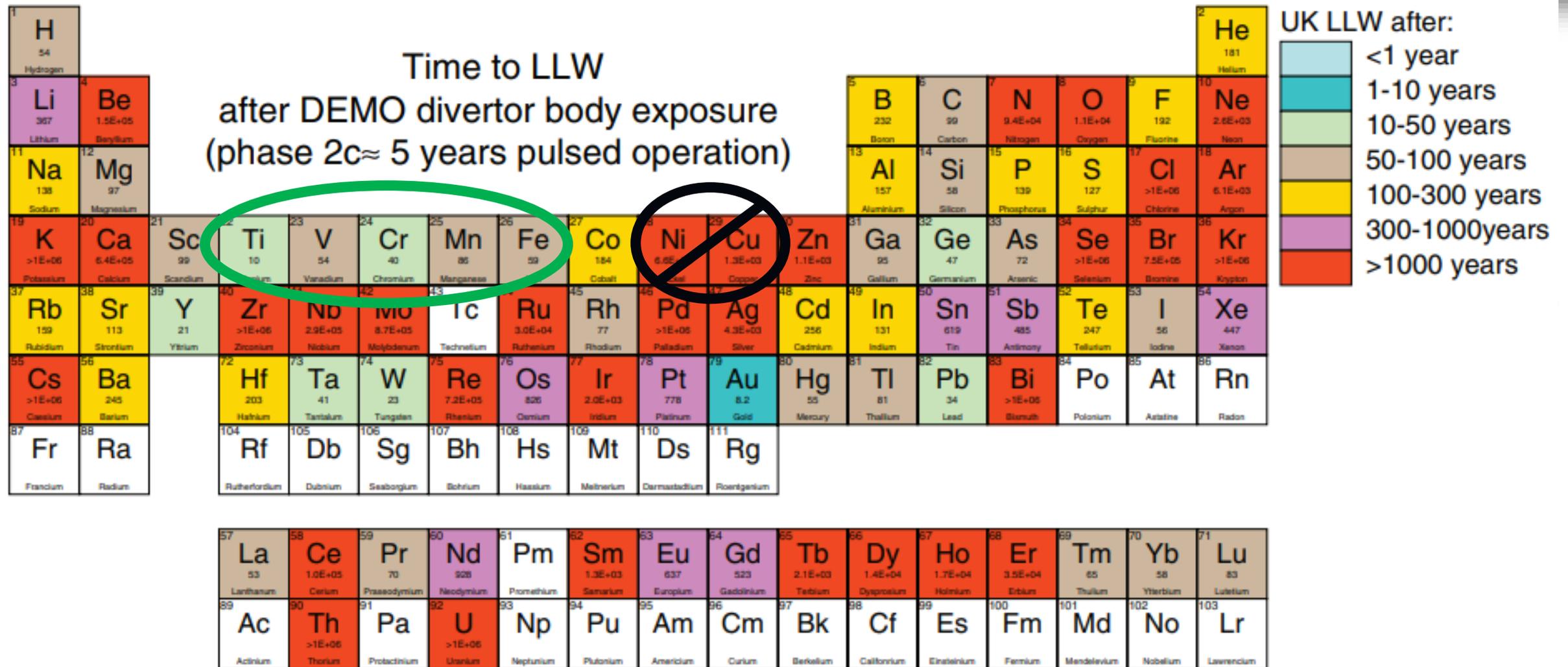
RADIATION TOLERANCE IN HEAs

Proposed mechanisms:

- Greater phonon scattering
- Sluggish diffusion of:
 - Atoms and point defects
 - Interstitial loops
 - ⋮
- More trapping sites
- Broad migration energies
- And more!



WHICH HEA COMPOSITION TO USE FOR ACCELERATORS?



Predicted cooldown time for elements used in DEMO reactor divertor environment (Gilbert | 2019)

WHICH HEA COMPOSITION TO USE FOR ACCELERATORS?

- Most commonly studied HEAs:

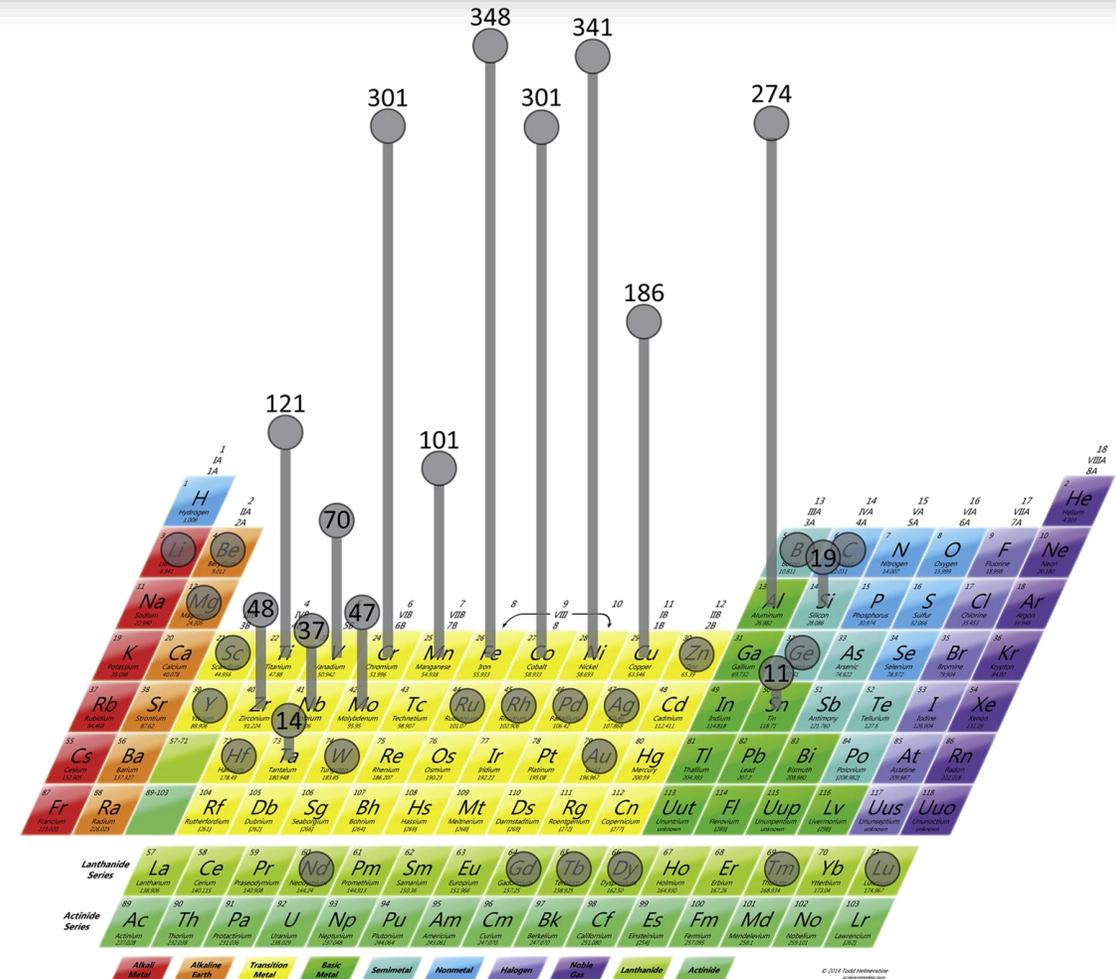
~~CoCrFeMnNi (Cantor alloy)~~

~~Al_xCoCrFeNi~~

~~Al_xCoCrCuFeNi~~

- New HEA using Cr, ~~Fe~~, Mn, Ti, V

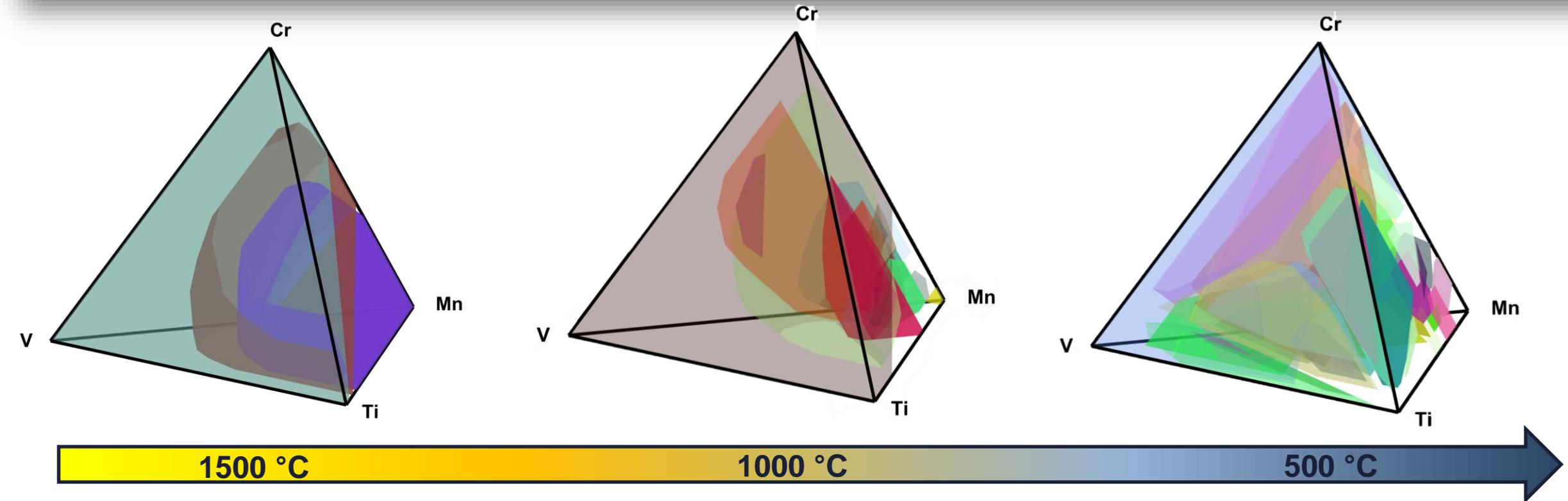
- Fe forms intermetallic compounds near the equimolar ratio with Cr, Ti, and V



Most commonly used elements in HEAs from literature (Miracle | 2017)



WHICH HEA COMPOSITION TO USE: CALPHAD



Takeaways:

Mn reduces the melting point substantially

V greatly promotes the formation of a single BCC phase

At lower temperatures, high **Ti** and **Mn** lead to the formation of many secondary phases



STARTING FROM LITERATURE



Contents lists available at [ScienceDirect](#)

Scripta Materialia

journal homepage: www.elsevier.com/locate/scriptamat



Towards V-based high-entropy alloys for nuclear fusion applications

P.J. Barron^{a,*}, A.W. Carruthers^a, J.W. Fellowes^a, N.G. Jones^b, H. Dawson^c, E.J. Pickering^a

^a Department of Materials, University of Manchester, Oxford Road, Manchester M13 9PL, UK
^b Department of Materials Science and Metallurgy, University of Cambridge, 27 Charles Babbage Road, Cambridge CB3 0FS, UK
^c Culham Centre for Fusion Energy, Abingdon OX14 3DB, UK



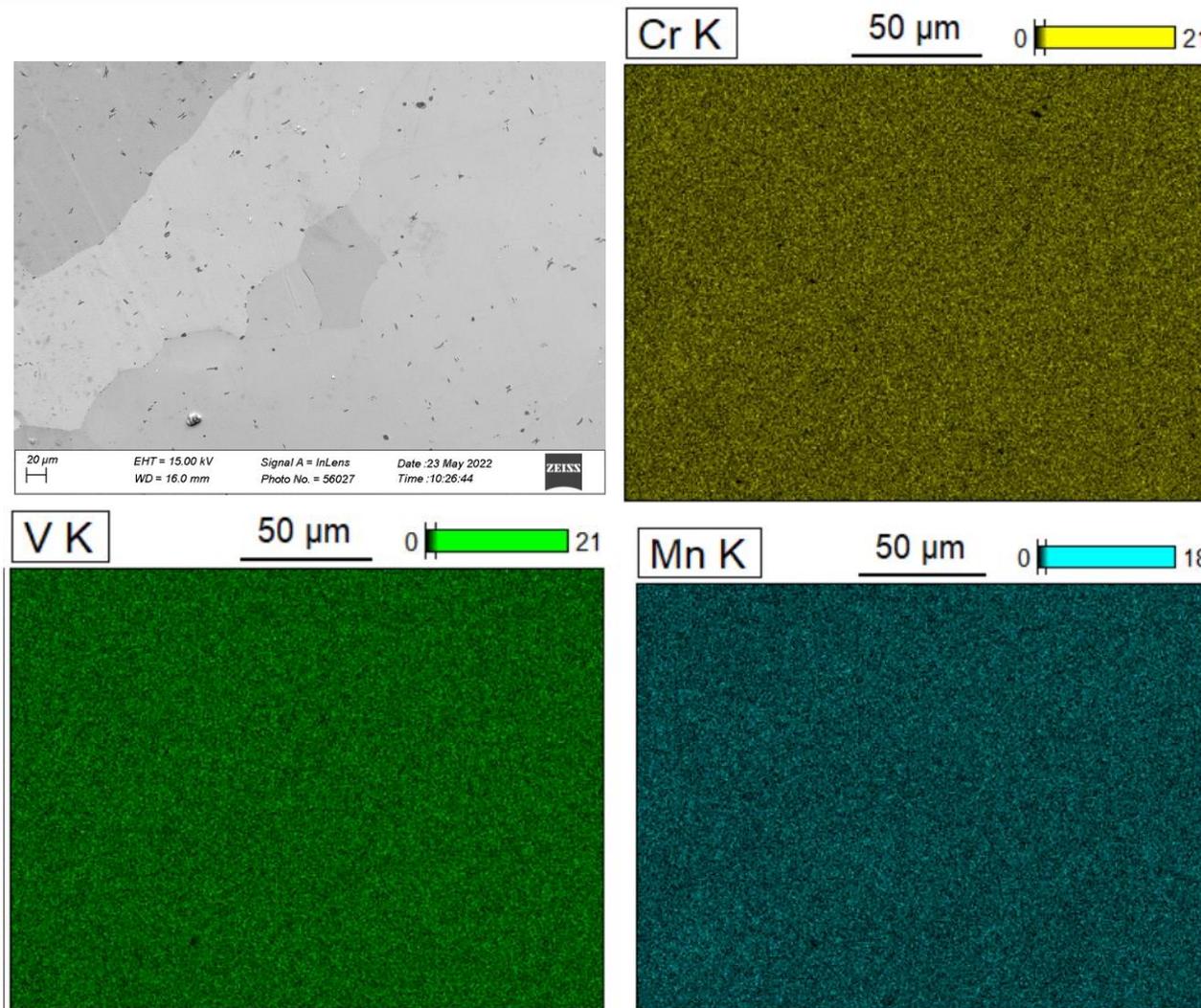
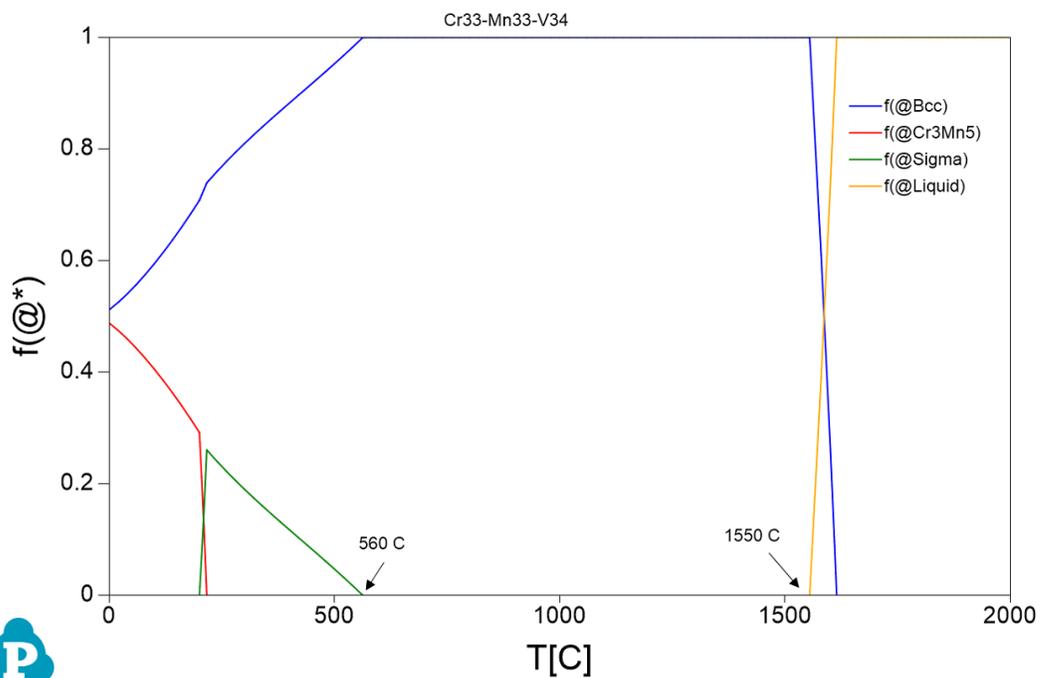
- Equimolar Cr-Mn-V shown to be single-phase BCC
- **Ti additions reduce concentration of interstitial impurities** by forming Ti-(C,O,N)
- Concerns **Ti concentrations above ~8% will promote Laves phase formation**
- 1200-°C, 100-hour heat treatment followed by water cooling to homogenize



$\text{Cr}_{33}\text{Mn}_{33}\text{V}_{33}$

$\text{Cr}_{33}\text{Mn}_{33}\text{V}_{33}$

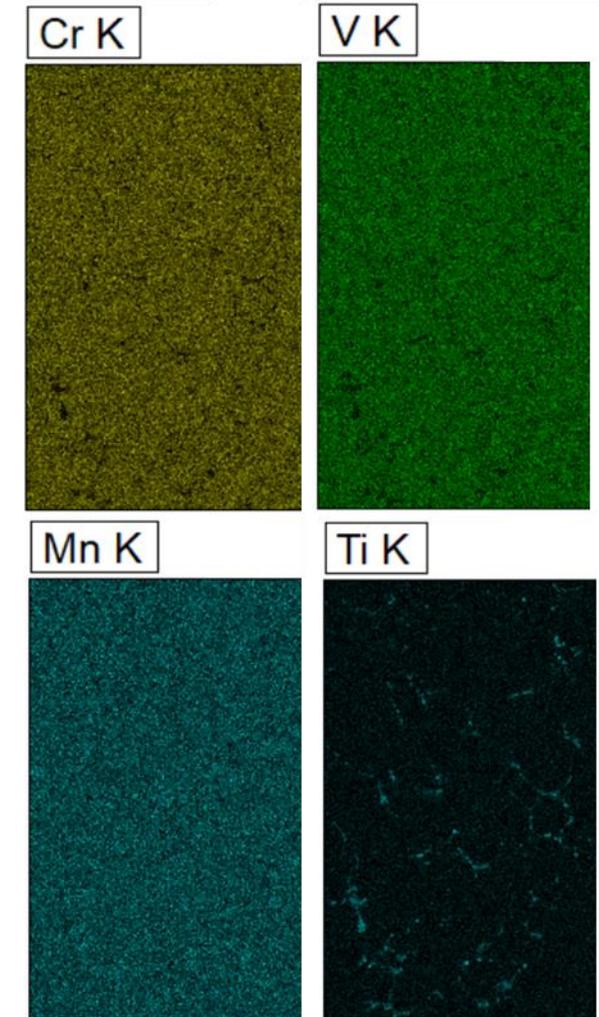
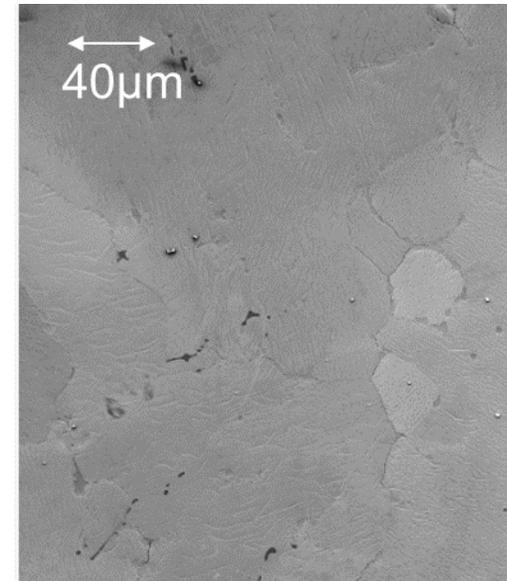
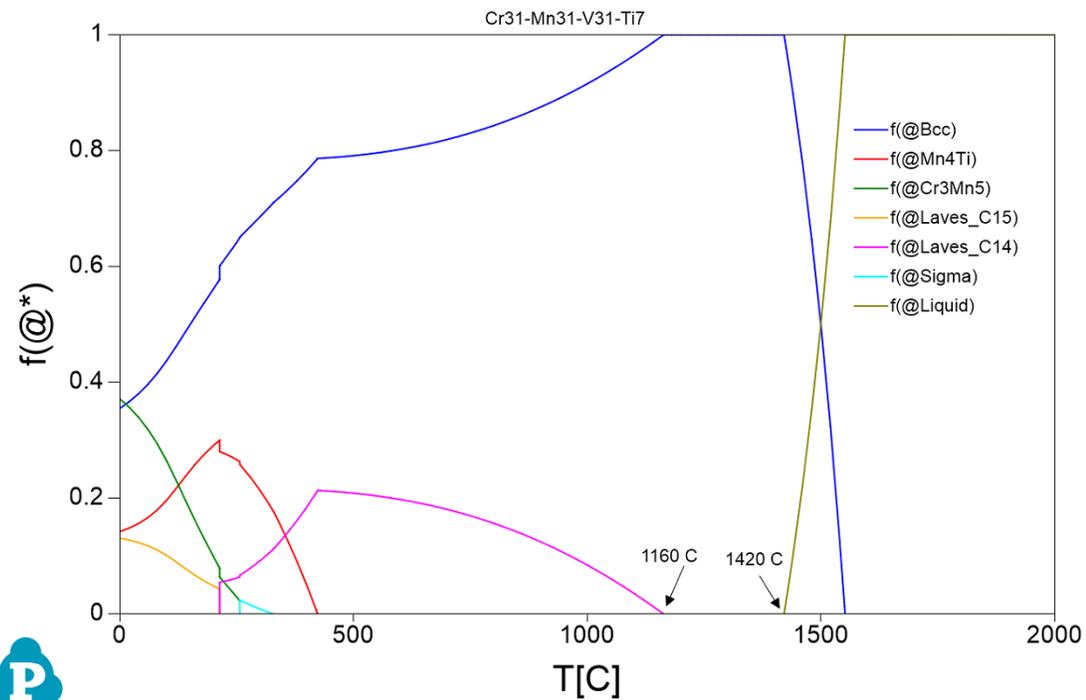
- Broad BCC phase field spanning ~1000 °C
- At 500 °C, the alloy is still >90% BCC





$\text{Cr}_{31}\text{Mn}_{31}\text{V}_{31}\text{Ti}_7$

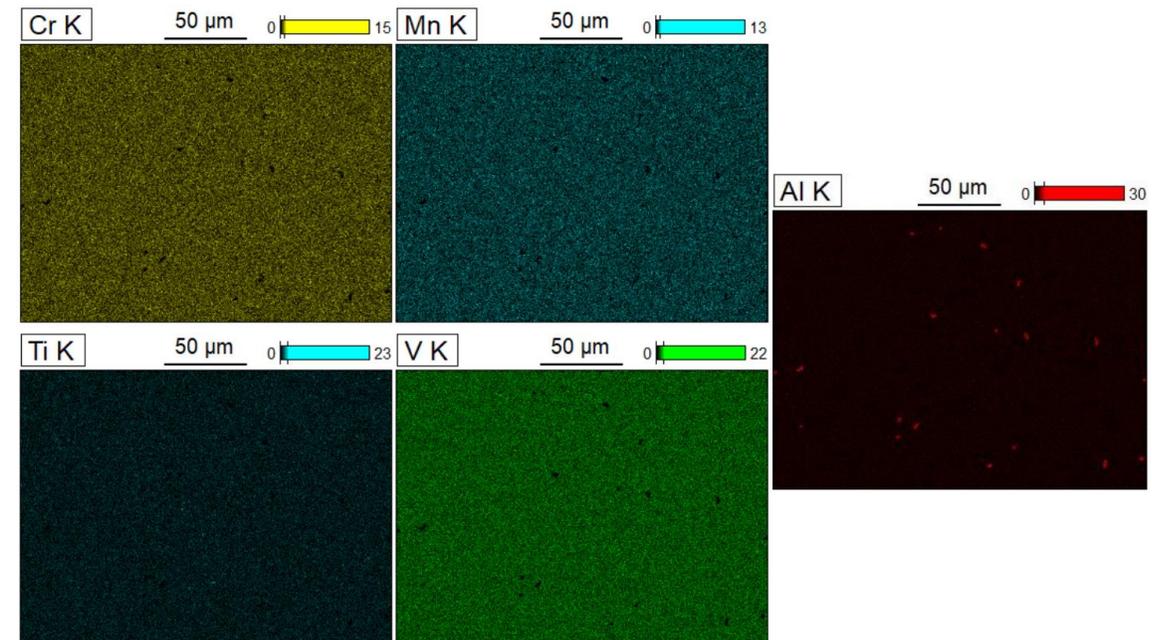
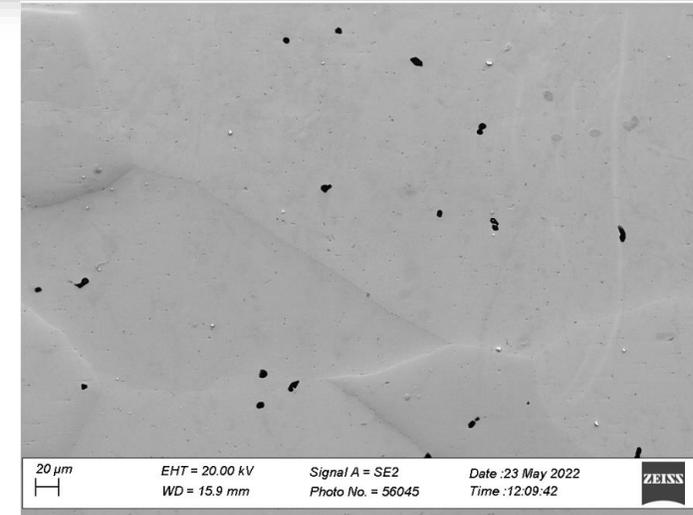
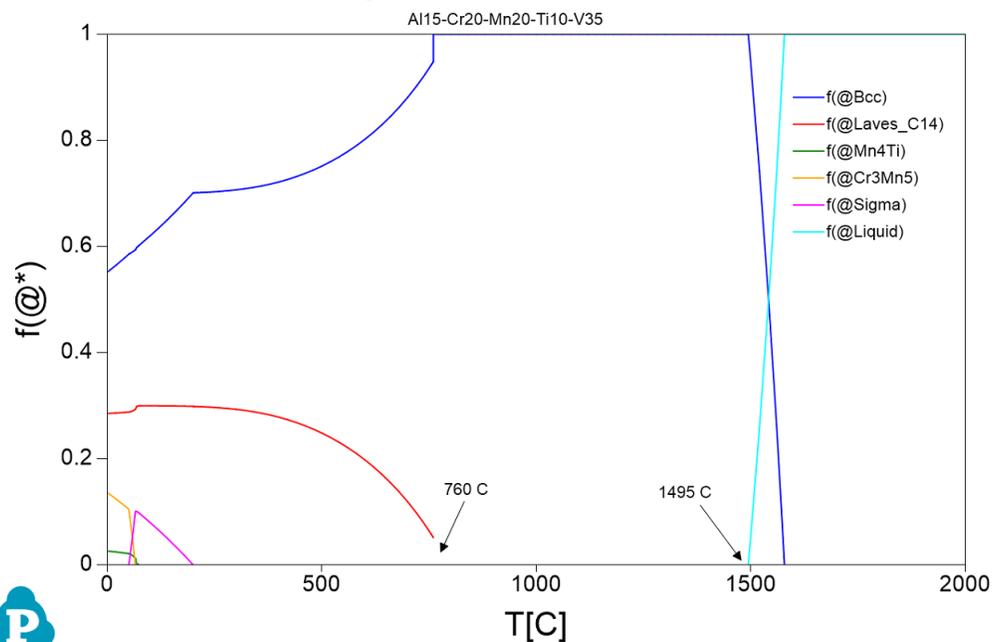
- Narrow BCC phase field spanning $\sim 260^\circ\text{C}$
- Laves phase (undesirable) predicted to form below 1160°C



AL-CR-MN-TI-V HEA



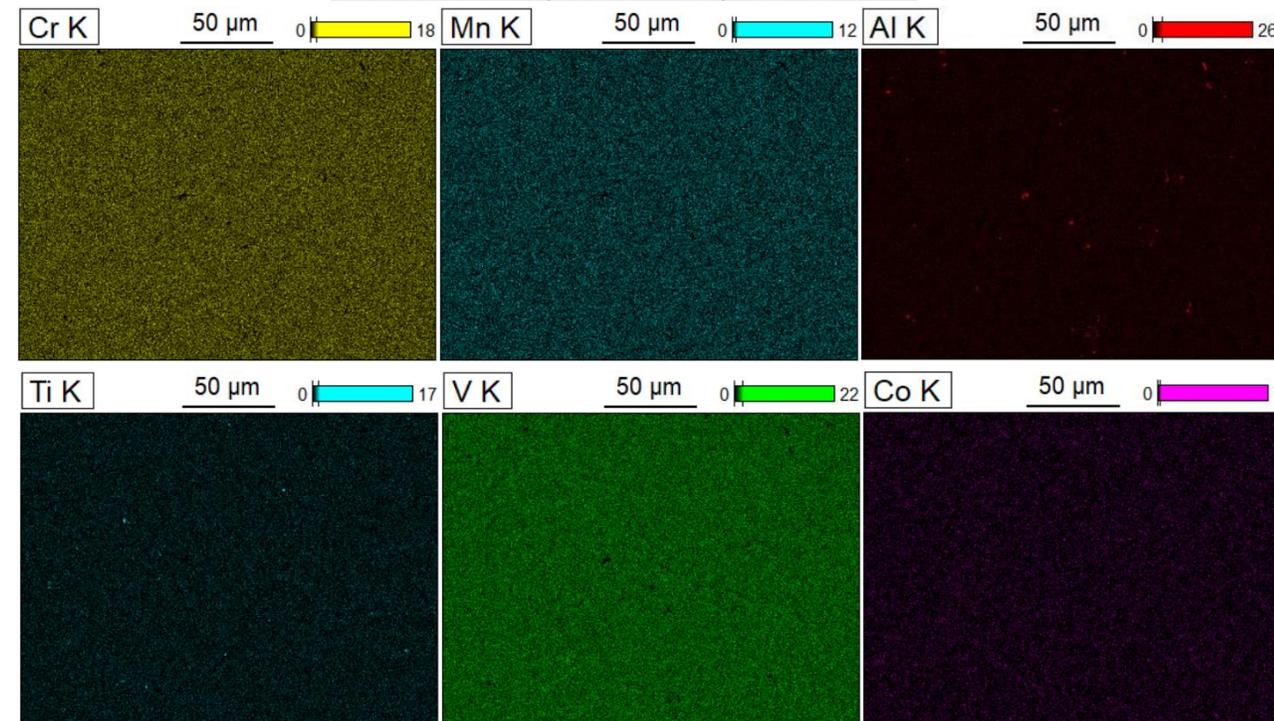
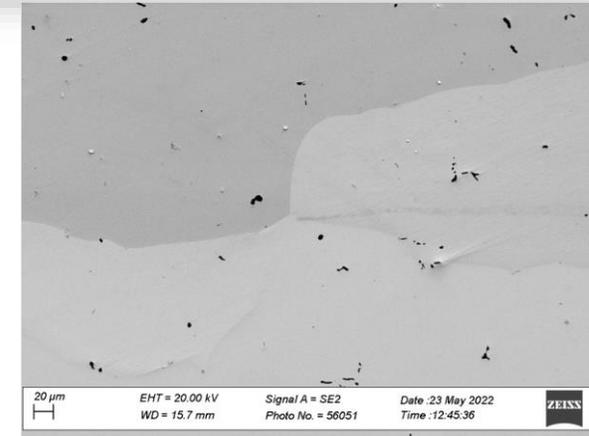
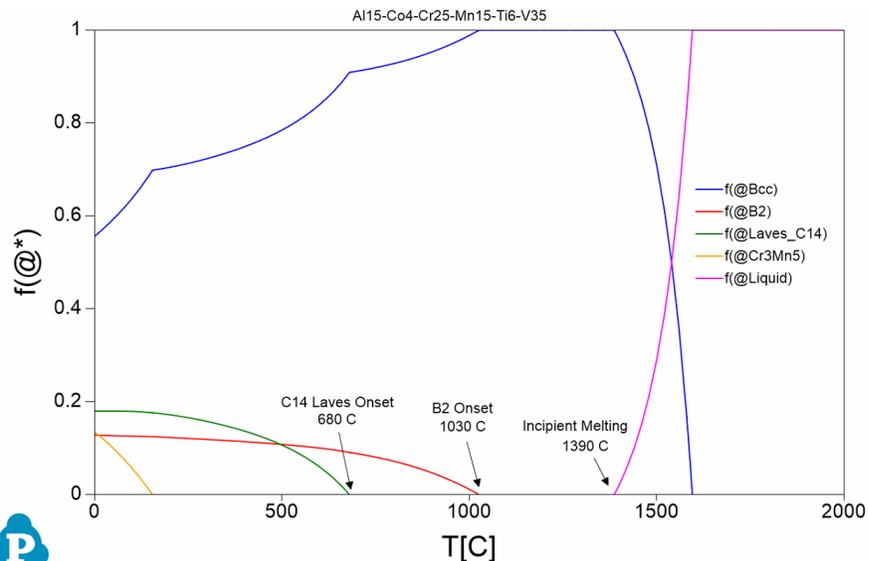
- Al shown to be a BCC stabilizer in multicomponent alloys + light weight
- Expands BCC stability range (ΔT) to 735 °C
- Increases melting point 75 °C



AL-CO-CR-MN-TI-V HEA

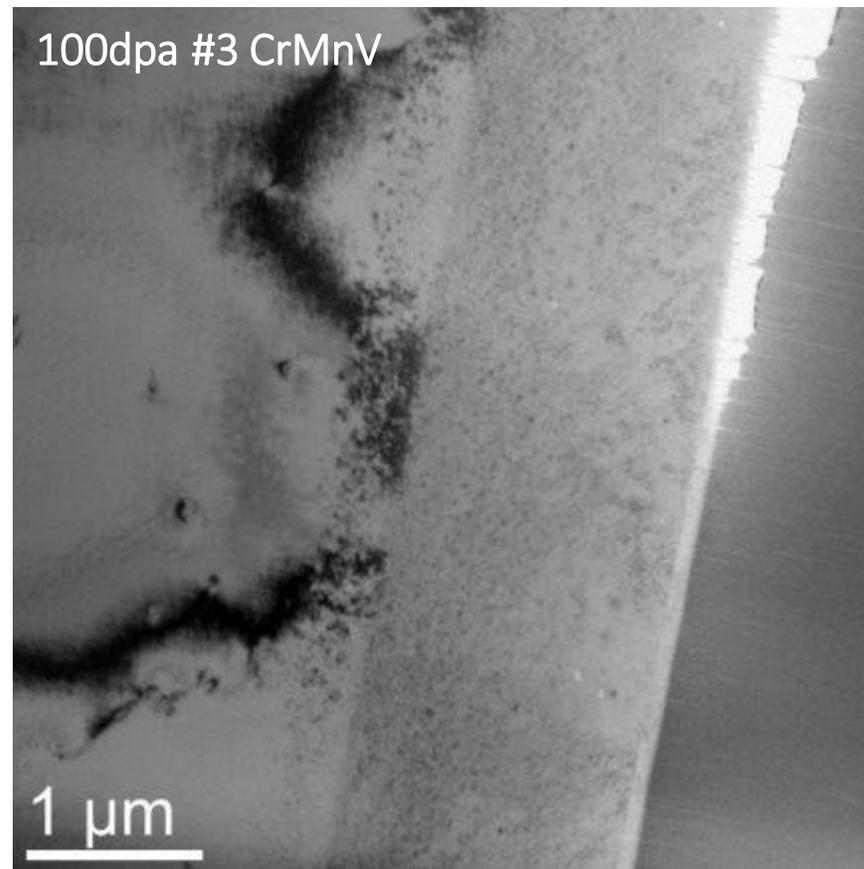
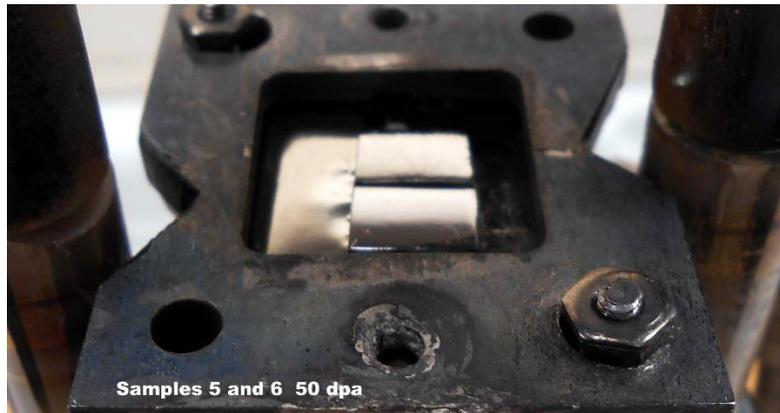


- 4% added Co at the cost of Ti promotes moderate B2 phase formation while maintaining sufficient Ti for impurity getting
- Further substitution of Cr for Mn helps to mitigate solidus temperature depression from Co additions



V^{2+} ION IRRADIATION 50/100 DPA AT 500C

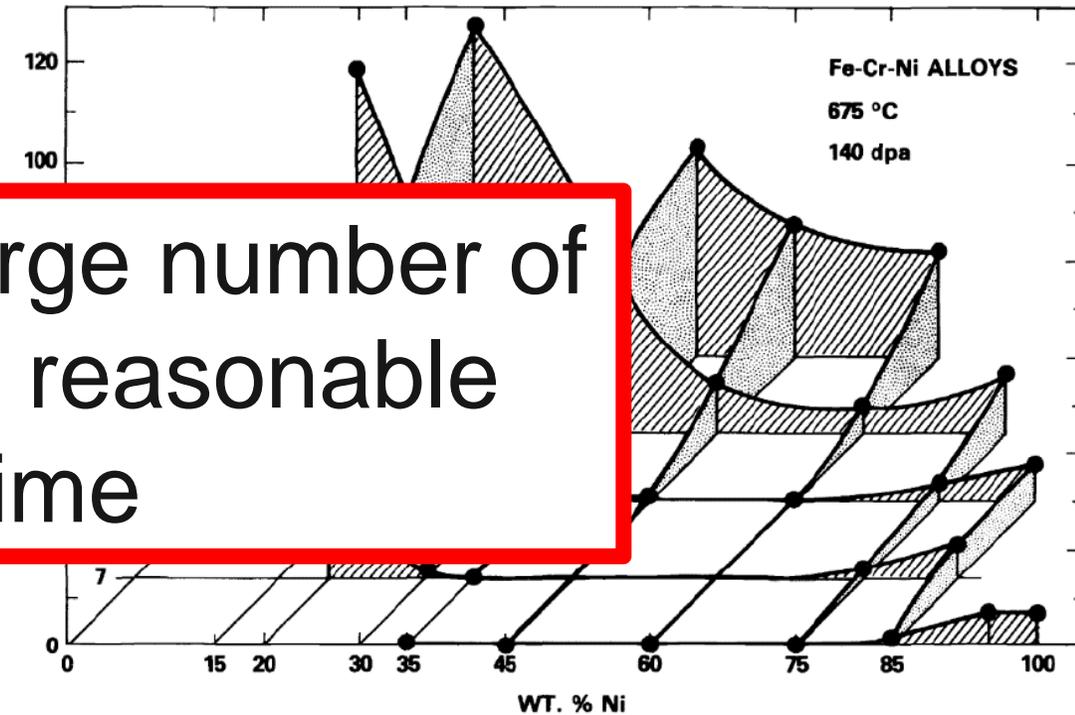
- PIE is ongoing



UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

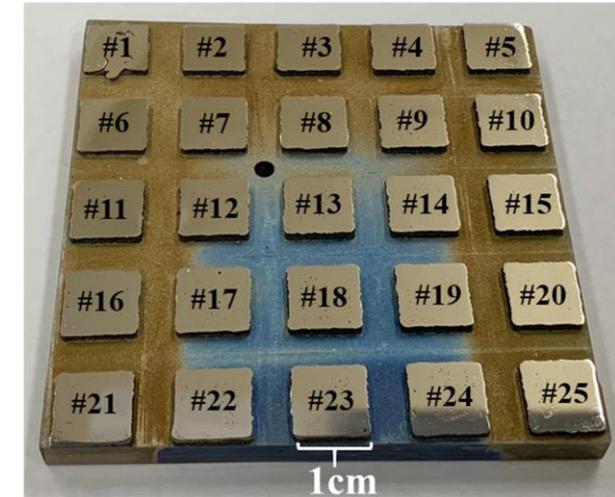
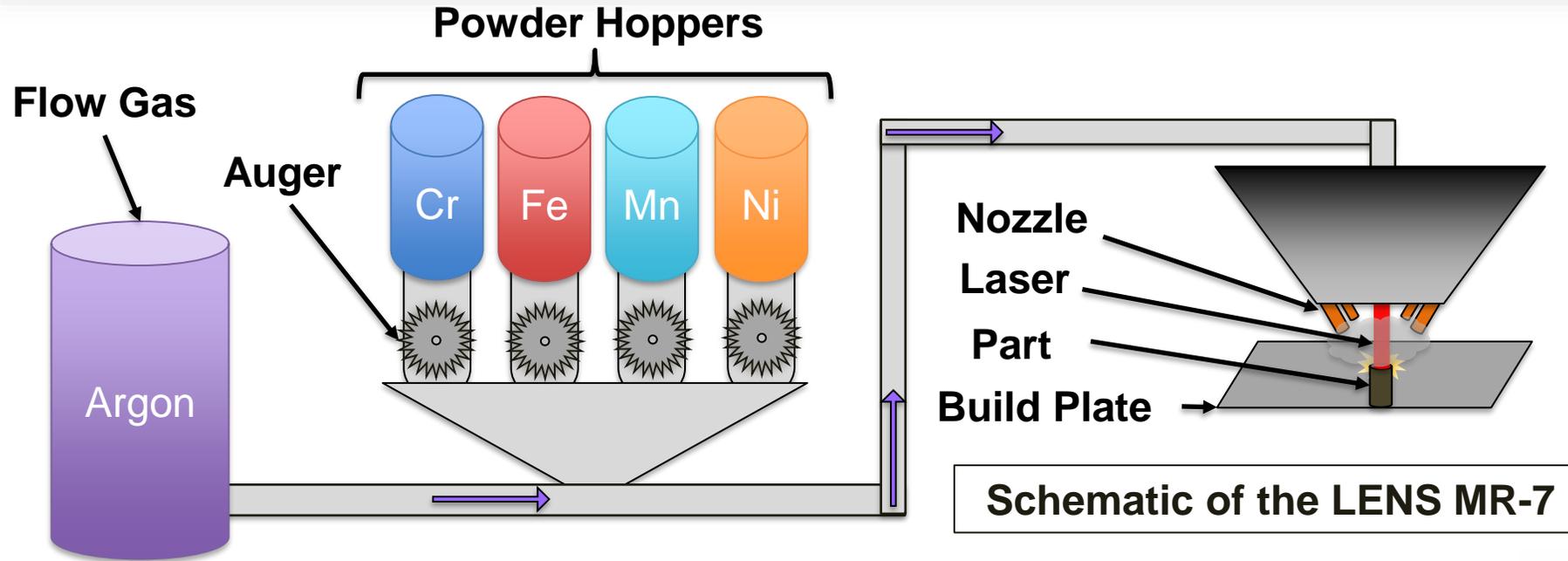
- Metallurgic study of large number of alloys is both expensive and time consuming
- Study of n is expensive require sp
- Standard ion-irradiated materials study time increases linearly with number of alloys

Objective: Study a large number of different alloys in a reasonable period of time



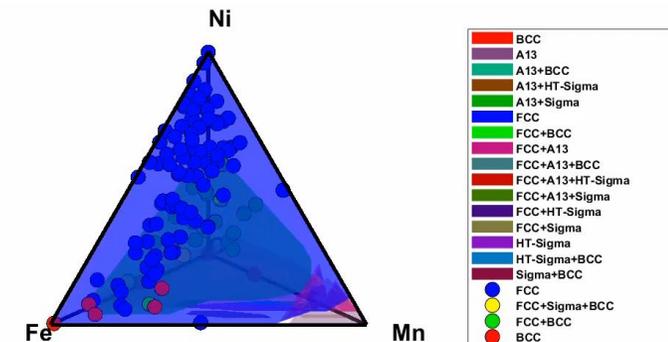
Johnston et al., 1982

UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs



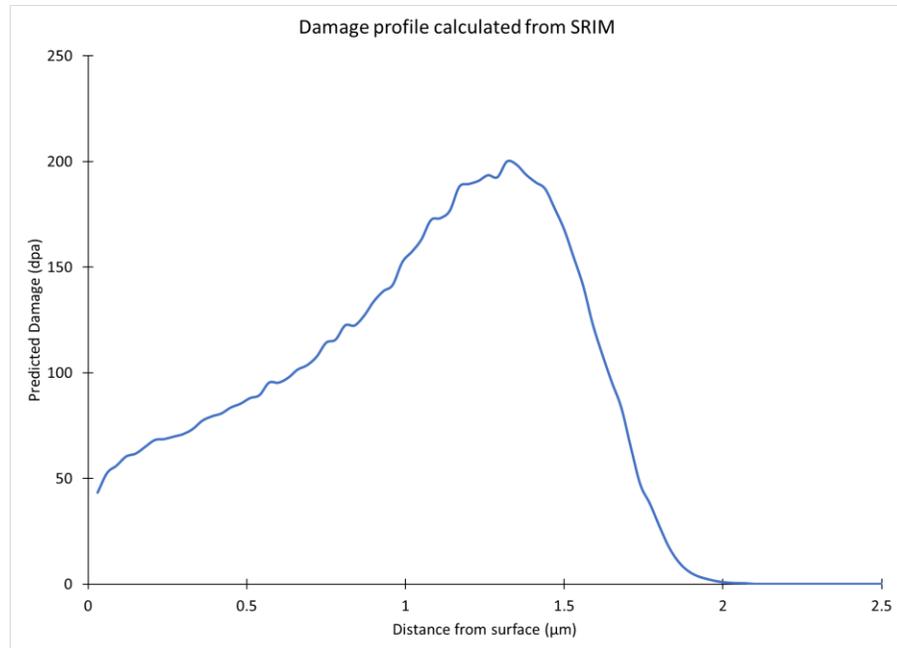
IN COLLABORATION WITH PROF. DAN THOMA AT UW-MADISON

- Elemental powders are controlled independently
- Powders are delivered to print head by argon flow gas
- Laser down optic axis melts powders
- Focus on steels component FeCrMnNi for method validation

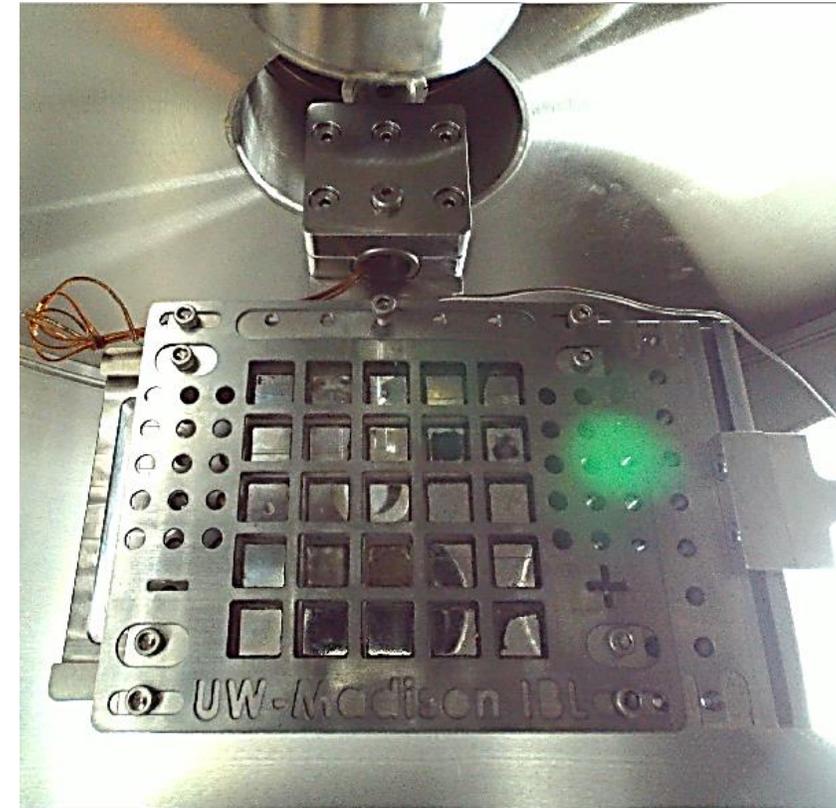


UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

- 4 MeV Ni²⁺ ions
- High temperature: 500 °C
- 200 dpa at peak damage



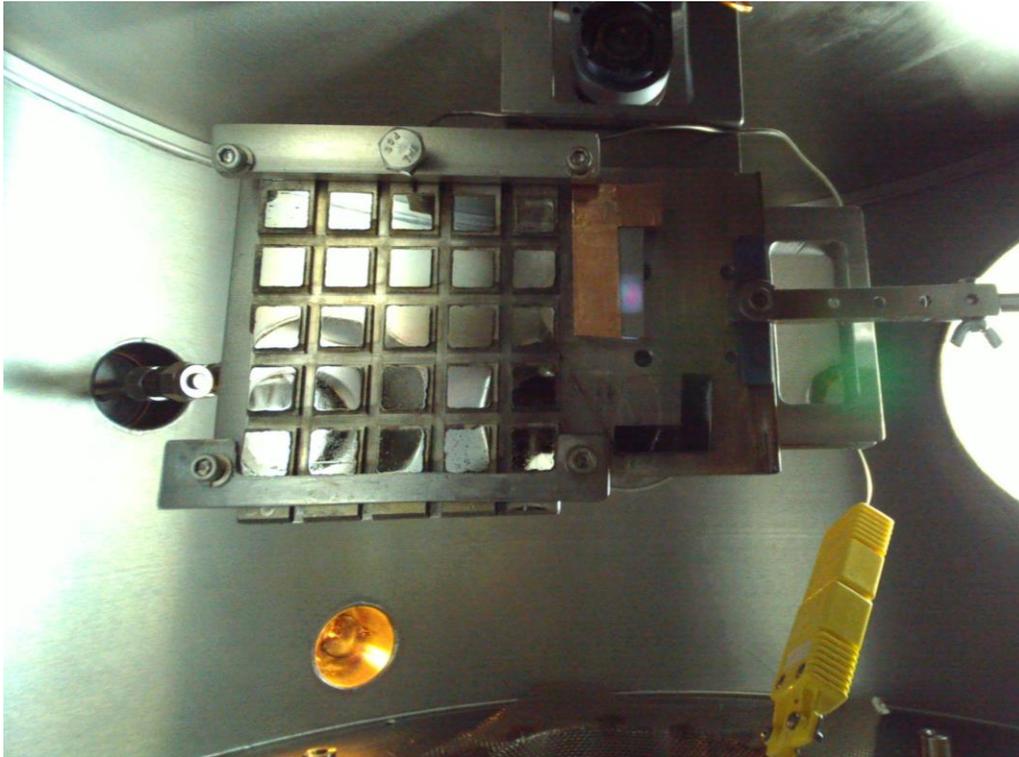
Challenge: Irradiate 25 samples in a limited period of time



Development of the high-throughput irradiation stage and chamber

UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

- Necessity to shape the ion beam to irradiated only half of each sample → Comparison between irradiated and un-irradiated areas



HEA additively printed plate in chamber



Ion beam positioning using electrostatic deflectors

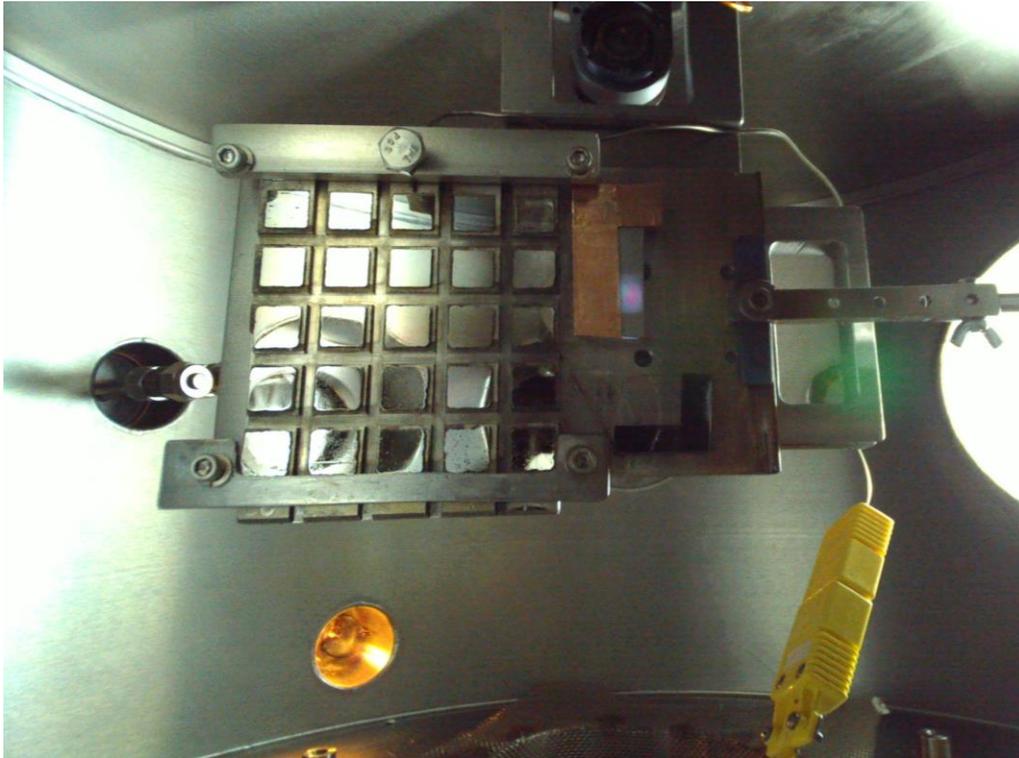


Ion beam shaping (Ta apertures) and Automated Faraday cup insertion for current reading



UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

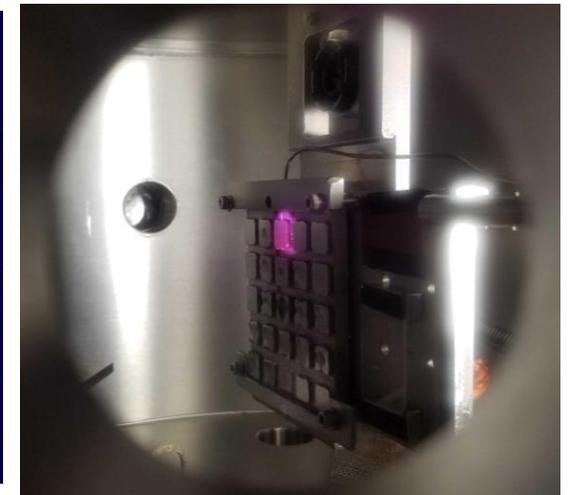
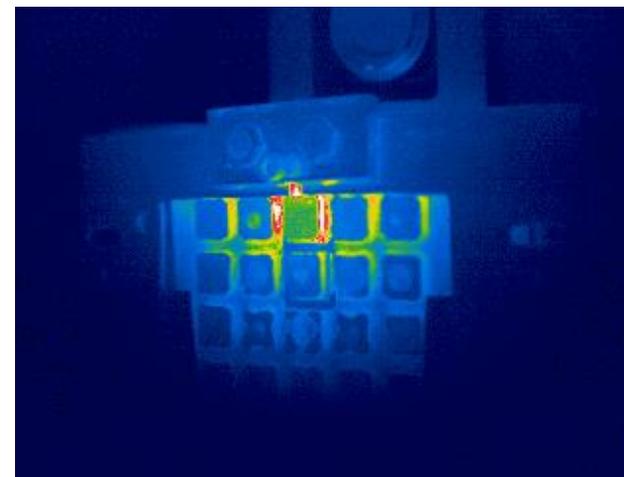
- Need to perfectly align three beams (ion beam, IR laser, pyrometer) on the same spot → Camera inside the irradiation chamber



HEA additively printed plate in chamber



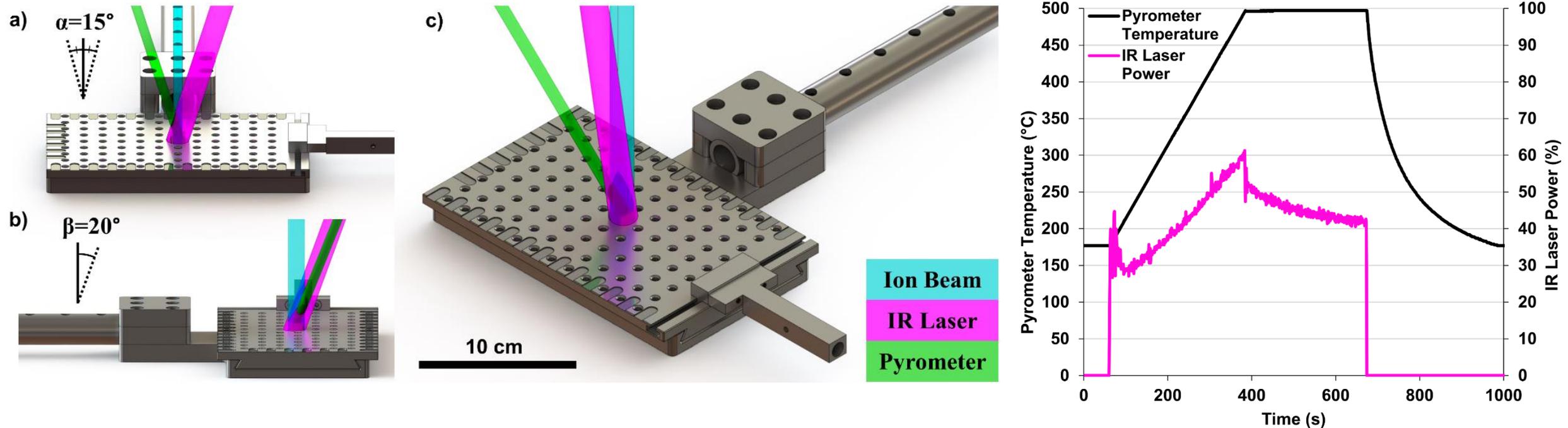
2kW laser turning off and still glowing red hot while the surrounding samples are cool



In-situ IR camera and pyrometer reading for temperature control



UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs



- IR laser (2kW) to heat the sample surface during irradiation (irradiation at temperature)
- Pyrometer to detect temperature in-situ and feedback loop to laser power
- Can reach 500C in 316SS in minutes with excellent temperature control

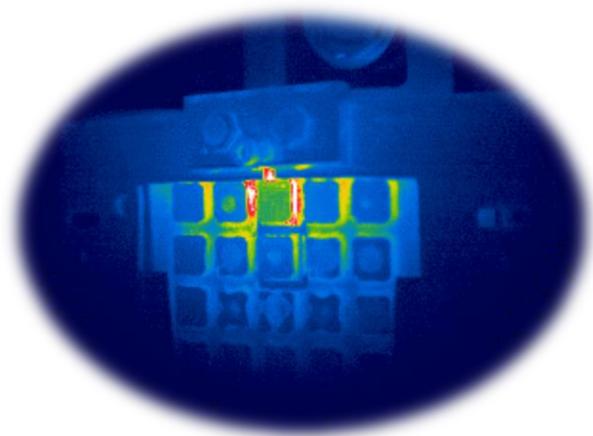
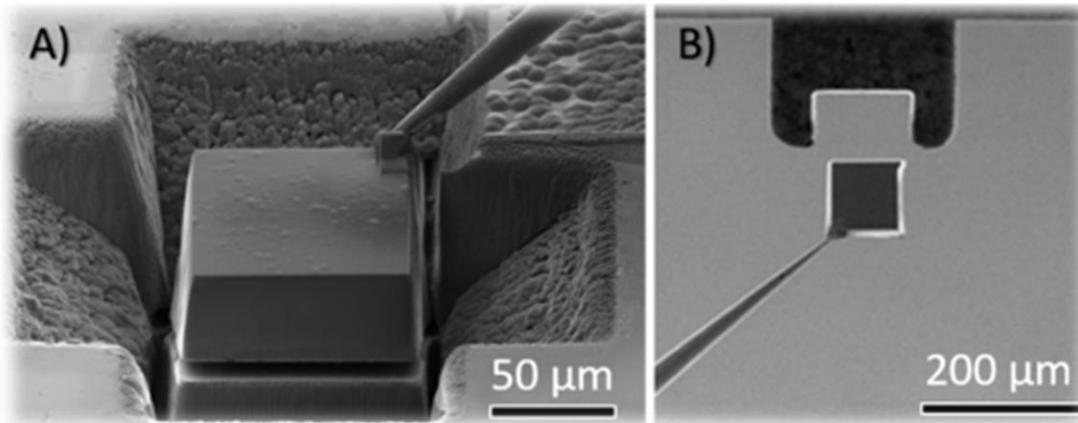
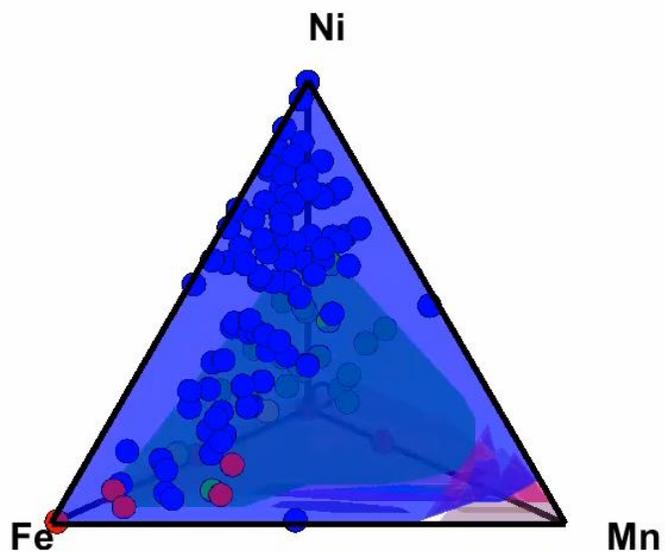
M. Moorehead, B. Queylat, H. Zhang, K. Kriewaldt, A. Couet, Development of a versatile, high-temperature, high-throughput ion irradiation system, Nucl. Instruments Methods Phys. Res. Sect. A 1020 (2021) 165892.

UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

Number of samples	75
Ions	4 MeV Ni ²⁺
Irradiation temperature	500 °C
Vacuum	10 ⁻⁶ torrs
Peak damage	200 dpa
Damage rate	0.04 dpa.s ⁻¹
Total irradiation time	100 hours



MADCOR



WISCONSIN
UNIVERSITY OF WISCONSIN-MADISON



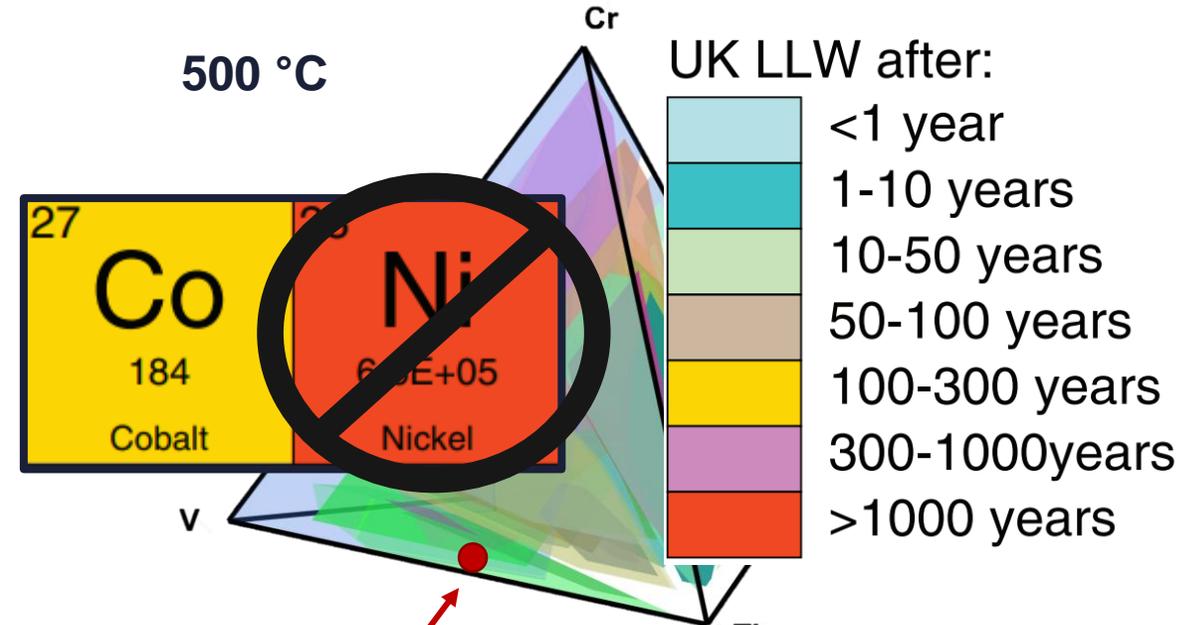
UW – ION BEAM LABORATORY (IBL)

- TORVIS: High current H, D, and He ions, from $< 1 \mu\text{A}$ to $100 \mu\text{A}$. Large flat damage/depth profile
- SNICS: Wide range of heavy ions, Fe, Si, C, V, Nd, and more. Fast to achieve high damage, e.g., 250 dpa peak damage of stainless steels in less than 8 hours. No radiation-induced radioactivity.
- Rastered or defocused beam can be provided

Beamline	Irradiation damage	Surface analysis	High-throughput irradiation
Temp Range:	$50^{\circ}\text{C}-1000^{\circ}\text{C} \pm 30^{\circ}\text{C}$	$-150^{\circ}\text{C}-1100^{\circ}\text{C} \pm 5^{\circ}\text{C}$	RT to 800°C (laser heating)
Temp. control	2 thermocouples + IR camera	3 thermocouples + IR camera	thermocouples + IR camera
Flux Range (cm ²)	$5 \times 10^{12} - 2 \times 10^{14}$	$1 \times 10^{10} - 2 \times 10^{15}$	$1 \times 10^{10} - 2 \times 10^{15}$
Irradiation Area	1.5 - 2.3 cm ²	0.1 - 6cm ²	0.1 - 225cm ²
Vacuum	1e-7 Torr	1e-8 Torr	5e-7 Torr
In-situ analysis	Resistivity measurement	RBS,NRA,PIXE	3D movement and rotation, and variable heating
Sample size	1cm x 2cm	1cm x 1.5 cm	15cm x 15 cm

STRETCH GOAL: ADD SEMI-COHERENT PRECIPITATES

- Ordered BCC phase (B2) expected to be semi-coherent with BCC lattice
- No Mn-containing (or concentrated) CrMnTiV alloys exhibit a B2 phase at low temperatures
- Dilute Ni or Co additions to $Al_{15}Cr_{20}Mn_{20}Ti_{10}V_{35}$ promote B2 formation



Predicted cooldown time for elements used in DEMO reactor divertor environment (Gilbert | 2019)

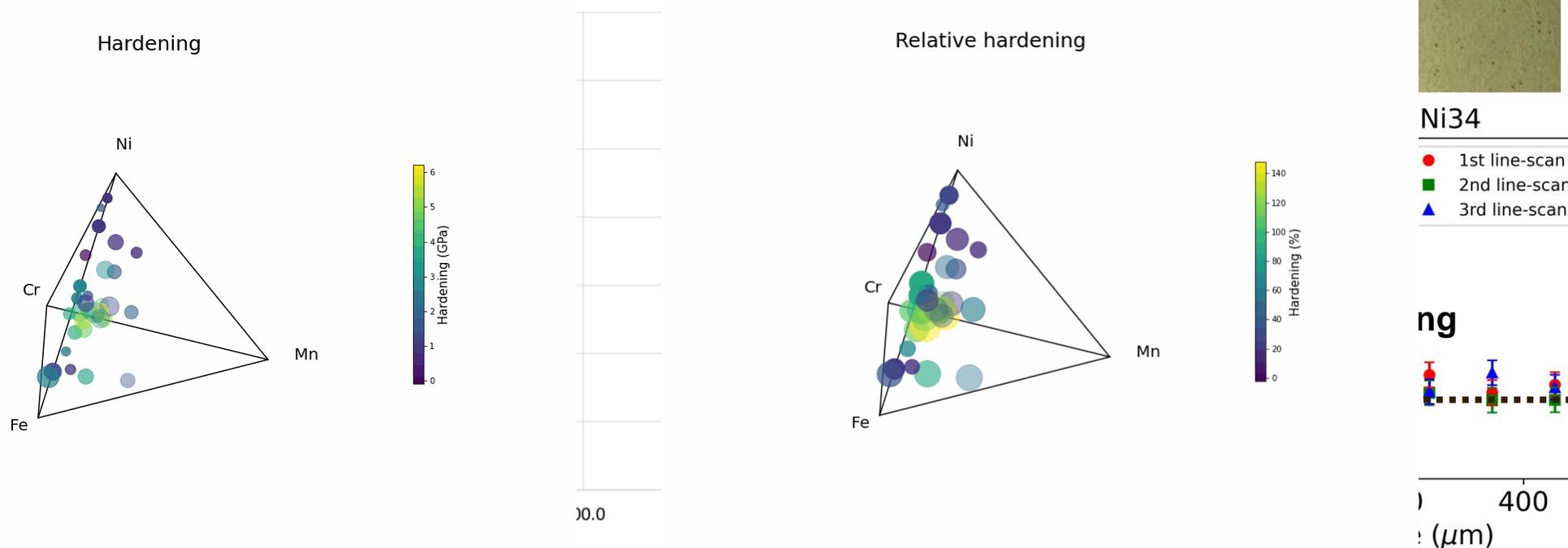
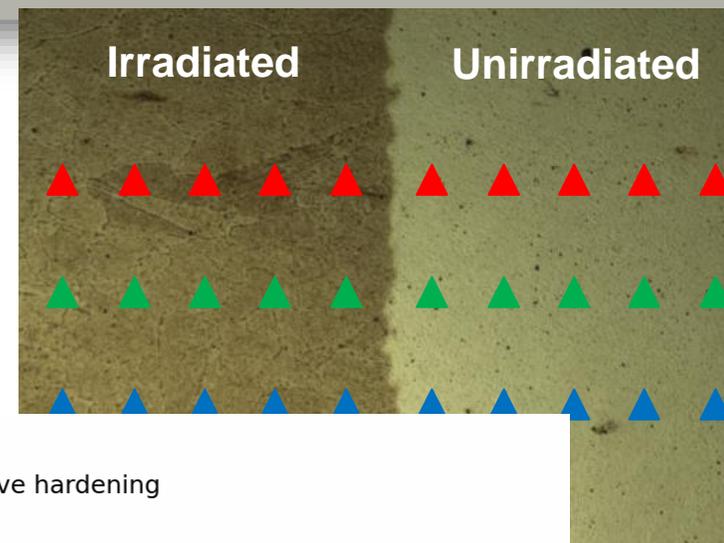
Alpha _i Mn+Laves _c 14	Bcc+Hcp+Laves _c 14+Laves _c 15	Bcc+Sigma+Cr3Mn5	Hcp+Bcc#1+Bcc#2+Laves _c 14
BCC _B 2+Laves _c 15	Bcc+Hcp+Laves _c 15	Bcc+Sigma+Laves _c 14	Hcp+Beta _i Mn
Bcc	Bcc+Hcp+Mn4Ti	Bcc+Sigma+Laves _c 14+Cr3Mn5	Hcp+Laves _c 14
Bcc#1+Bcc#2	Bcc+Hcp+Mn4Ti+Laves _c 14	Beta _i Mn+Laves _c 14	Hcp+Laves _c 14+Laves _c 15
Bcc#1+Bcc#2+Laves _c 14	Bcc+Laves _c 14	CBCC _A 12	Hcp+Laves _c 15
Bcc#1+Bcc#2+Mn4Ti	Bcc+Laves _c 14+Cr3Mn5	CBCC _A 12+Cr3Mn5	Hcp+Mn4Ti
Bcc#1+Bcc#2+Mn4Ti+Laves _c 14	Bcc+Laves _c 14+Laves _c 15	Hcp	Hcp+Mn4Ti+Beta _i Mn
Bcc+BCC _B 2	Bcc+Laves _c 15	Hcp+Alpha _i Mn	Hcp+Mn4Ti+Laves _c 14
Bcc+Cr3Mn5	Bcc+Mn4Ti	Hcp+Alpha _i Mn+Laves _c 14	Laves _c 14
Bcc+Hcp	Bcc+Mn4Ti+Laves _c 14	Hcp+BCC _B 2	Laves _c 14+CBCC _A 12
Bcc+Hcp+BCC _B 2	Bcc+Mn4Ti+Sigma	Hcp+BCC _B 2+Laves _c 15	
Bcc+Hcp+BCC _B 2+Laves _c 15	Bcc+Mn4Ti+Sigma+Laves _c 14	Hcp+Bcc#1+Bcc#2	



UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

Irradiation induced hardening

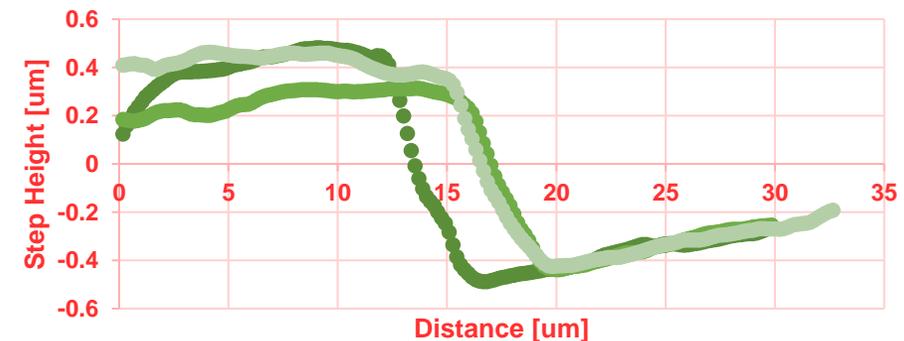
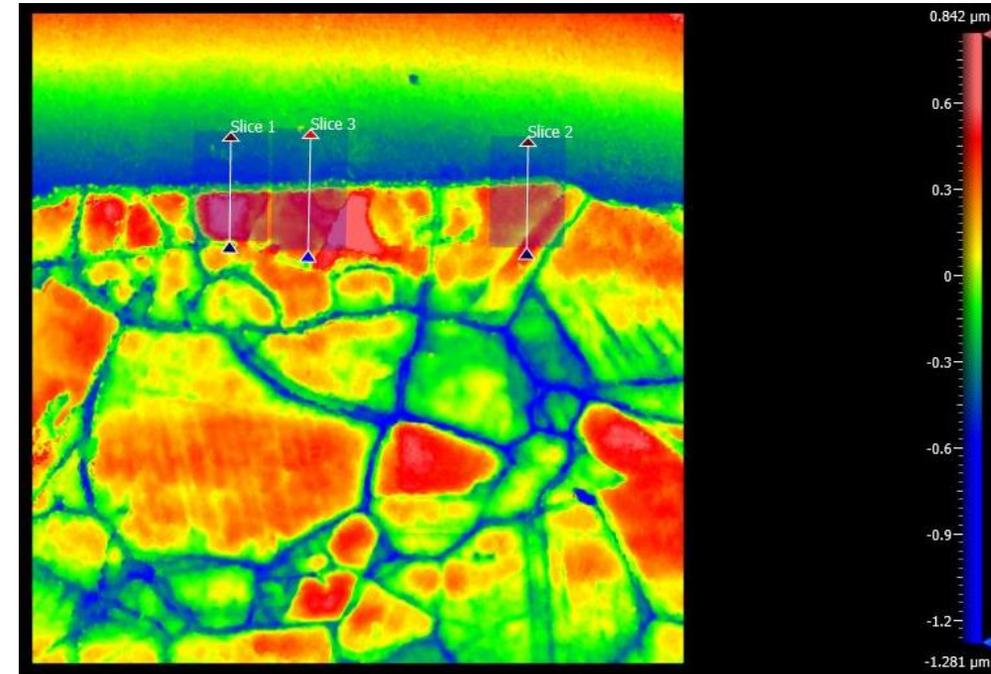
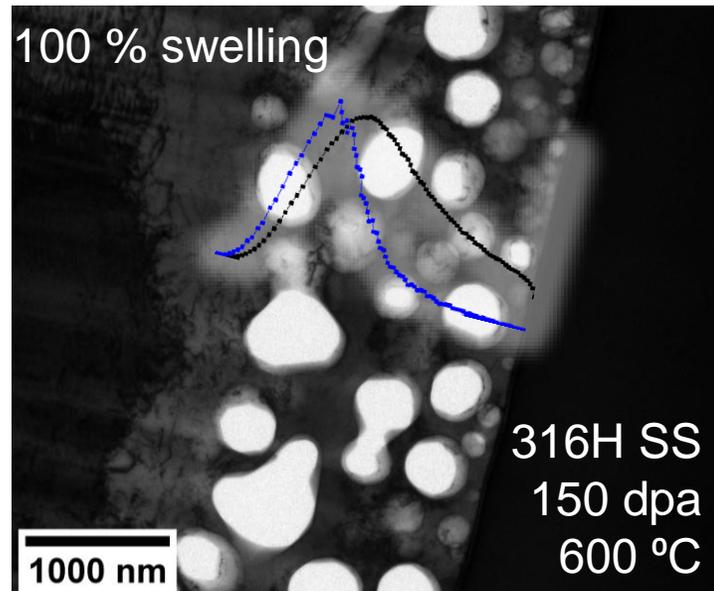
- For each samples: 3 line-scans (10 indents each) have been done across the interface
- Total time: 44 hours per plate (3mn/indent)



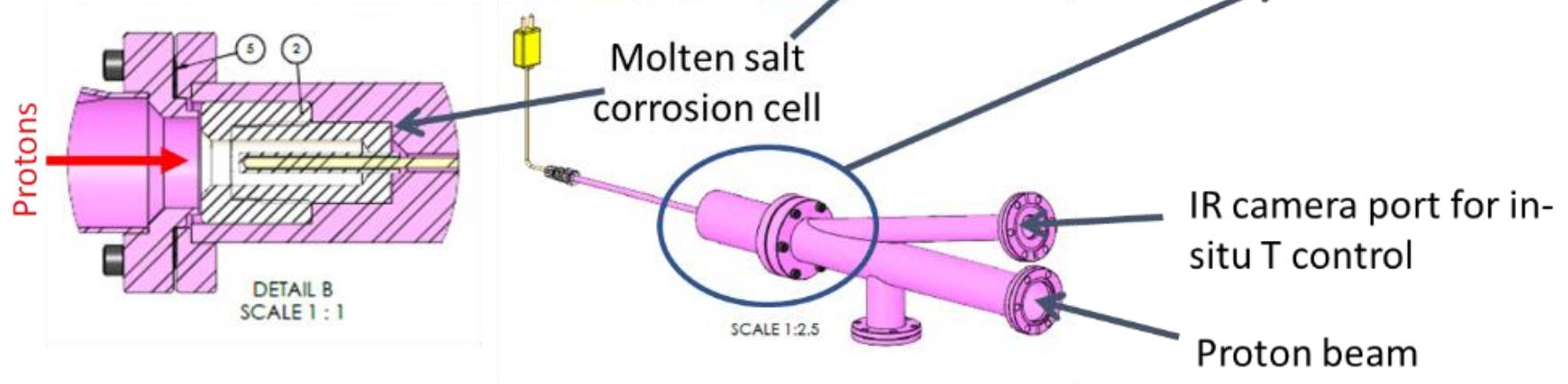
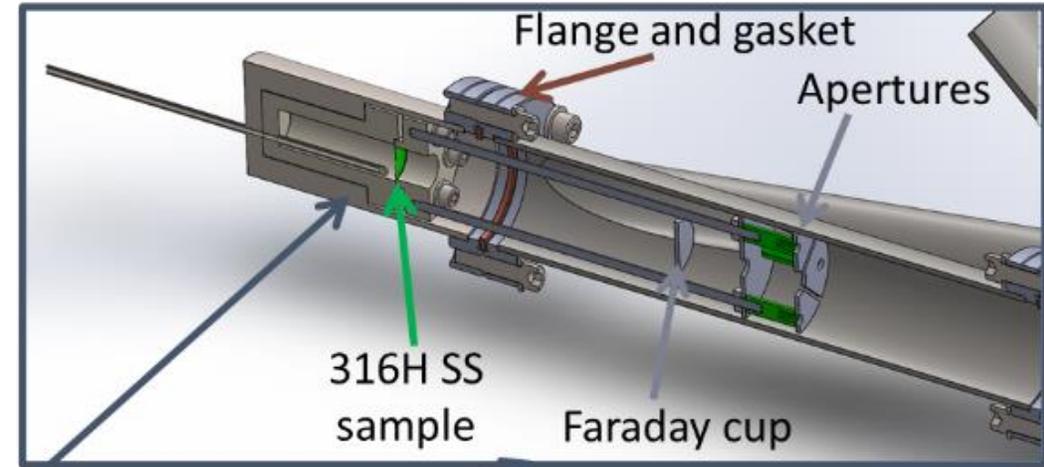
UW – IBL: EXAMPLE 2 – HIGH THROUGHPUT IRRADIATION OF HEAs

Irradiation induced void swelling measured by profilometry:

- Profilometry is a relevant semi-high throughput technique to measure irradiation induced void swelling.
- Under development for high-throughput capabilities



UW – IBL: EXAMPLE 3 – IN-SITU IRRADIATION IN MOLTEN SALTS (2022)



UW – IBL: POST IRRADIATION EXAMINATION (PIE) CAPABILITIES



UW – IBL: POST IRRADIATION EXAMINATION (PIE) CAPABILITIES

- For sample preparation of low-active materials
 - Parallel polisher
 - Low speed saw
 - Electro-polisher
 - Ion mill
 - High accuracy balance



UW – IBL: POST IRRADIATION EXAMINATION (PIE) CAPABILITIES

- JEOL 6610 SEM with Energy Dispersive Spectroscopy (EDS) and Electron Backscatter Diffraction (EBSD) capabilities in rad area:



UW – IBL: POST IRRADIATION EXAMINATION (PIE) CAPABILITIES

- For sample characterization in non-rad area (after sample prep in rad-area)



**FEI Helios G4
UX Plasma
FIB/FE SEM**



**FEI Titan Cs-corrected
scanning transmission
electron microscope**



**Hysitron TI 950
TribolIndenter**

